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Optical analysis of Hybrid quantum dots deposited by SILAR

method

Asmaa F. Mansour¹, Sawsan A. Mahmoud², and Moustafa E.Elsisi¹

¹Zagazig University, Physics department, Faculty of science, Zagazig, Egypt. ²Egyptian Petroleum Research Institute, postal code 11727, Nasr City, Cairo, Egypt. **Corresponding Author: Moustafa E.Elsisi;** <u>mostafaels999@gmail.com</u>

ABSTRACT: In the present study, ZnS and CdS quantum dots (QDs) are attached to mesoporous titanium film by successive ionic layer adsorption and reaction to the synthesized three layers of ZnS and CdS quantum dots (QDs) to form ZnS/TiO₂ nanocomposites (NCs), CdS/TiO₂, and CdS/ZnS/TiO₂ (NCs) as hybrid structures. TiO₂ nanoparticles (NPs) are fabricated by the sol-gel method. The synthesized nanomaterials were investigated by XRD, HR-TEM, FTIR, and UVVisible spectrophotometer. The size of TiO₂ NPs was investigated from XRD analysis to be 34.33 nm and was confirmed with HR-TEM. The UV–Vis absorption spectra for samples were recorded. The optical band gap energies (Eg) have been calculated for the deposited films and were found to be 2.39 eV, 2.30 eV, 1.98 eV, and 1.68 eV for TiO₂ NPs, ZnS/TiO₂, CdS/TiO₂, and/CdS/ZnS/TiO₂ NCs, respectively.

KEYWORDS: Sol-gel method; SILAR method; CdS QDs; TiO₂ NPs; Hybrid structures.

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I. INTRODUCTION

One-dimensional nanostructures, such as nanorods, nanowires, and nanotubes, exhibit interesting electrical and optical properties due to the quantum confinement effect and low dimensionality. Realizing the field of radiation with nano-science promotes wide opportunities in the field of photonics, radiation detectors, photo detectors, and in sensing applications [6-8]. Cadmium sulphide is the most essential II-VI n-type compound semiconductor because of its eminent properties such as nonlinear piezoelectricity, good thermal and chemical stability, optical behavior, direct band gap, and optical coefficient. Onedimensional CdS quantum dots contain several properties, such as a high aspect ratio, efficient transport properties, and so on, resulting in nextgeneration nanoscale electronic and photonic devices. One dimensional CdS quantum dots are distinguished materials for photovoltaic cells, nonlinear optics, optoelectronic devices, sensors, and photocatalytic applications [9-10]. Different methods, such as chemical bath deposition [15], spin coating [11], chemical vapor deposition (CVD) [12], and chemical dip coating [4] have been used to form semiconducting thin films. In the current work, CdS and ZnS thin films are deposited on the glass substrate using the SILAR method because it is a simple, efficient, and cost effective technique. The optical properties of semiconducting nanostructures can be modified by using a number of methods such as ion doping, irradiation, and ion implantation. The refinement in the optical properties such as the band gap is pretended due to the generation of impurity levels in the forbidden energy gap [3].

QDs are nanocrystalline materials that have a size in the range of 2 to 10 nm and have perfect photostability as well as large extinction coefficients [20]. Interestingly, QDs have received much awareness in the growth of solar cell technologies because of the quantum confinement effect that they display as their exciton Bohr radius decreases. A QD with a smaller size than its exciton Bohr radius is said to have strong custody, whereas a QD with a larger size than its exciton Bohr radius is said to have a weak quantum custody effect. QDs absorb and emit precise color at a specific spectral wavelength corresponding to their particle sizes. Generally, the smaller QDs emit blue light, while the largest nanosized QDs emit red light, with all the other colors appearing in between. This color-changing phenomenon indicates that QDs of a certain particle size take possession of specific optoelectronic properties and differ in terms of their bandgap energies. The bandgap is the required energy to raise the electrons from the (V.B) to (C.B). Theoretically, small-sized dots have a higher energy gap

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than their bulk counterparts and require energy equal to or higher than their bandgap energy to enter an excited state. The discipline of QD particle size, energy level, and emission wavelength makes them extremely useful for optoelectronic applications such as photovoltaic cells [16-18] and sensors [21-30]. The successive ionic layer adsorption and reaction (SILAR) is the preferred technique for the adsorption of CdS and ZnS QDs on TiO2 NPs. This work focuses on the processing of QDs based on the SILAR method, and optical properties of hybrid structures are discussed corresponding to their bandgap energy and spectral absorption [2]. Most research has focused on engineering the device structure [39, 40] and functionalizing QD surfaces to improve the performance of QD-based LEDs [41, 42, 43].

In our work, the effect of different quantum dots on the optical behavior of TiO2 NPs is further investigated, which is more importantly associated with ZnS over TiO2 QDs, CdS over TiO2 and hybrid structures of CdS and ZnS over TiO2. The optical energy gap of TiO2 was studied after the deposition of ZnS, CdS, and CdS/ZnS at 50 °C. 3- Environmental factors

II. MATERIALS AND METHODS

Apparatus

(Titanium (IV) isopropoxide (97%, Aldrich), Zinc Acetate Zn (CH3COO) 2 (97%, Bio Chem), Sodium Sulfide Na2S (98%, Alpha chemical), Cadmium Nitrate Cd (NO3)2 (97%, Bio Chem), Ethanol absolute (99%, Bio Chem), Triton -X100 (97%, Aldrich), (FTO conducting glass).

2.1. Synthesis of TiO2 nanoparticle by Sol-gel method:

TiO2 NPs were prepared by a sol-gel approach using Titanium (IV) isopropoxide (TIP) [Ti [OCH (CH3)2]4 as a precursor according to [8]. The sol-gel method includes the condensation and hydrolysis processes of titanium (IV) isopropoxide in aqueous media under acidic conditions. The procedure used is followed by preparing two solutions: In solution (1), TIP was added to absolute ethanol at a ratio of one to three by volume under continuous stirring for 30 minutes until a homogenous clear yellow solution formed. Solution (2) contains DI water and absolute ethanol at a volume ratio of one to four. Then, HNO3 as N2 atmosphere is mixed well into the solution by dropwise addition until the pH value approaches about 2 under continuous stirring for 1 h at room temperature to restrain the hydrolysis process of the solution. Finally, solution 2 was added slowly to solution 1 and aged under robust stirring for 2 h. The gel was digested at 80 °C in water path for 1 h until most of the ethanol evaporated, then dried for 24 h in a regular laboratory oven at 80°C and then calcined at 400 °C for 2 h [33].

2.2. Thin films preparation

A TiO2 NPs film with the same thickness of 0.0051g/cm2 was prepared. The samples of TiO2 NPs film were directly deposited onto glass substrates by the doctor-blading method. Before spreading the film, the glass sheet surfaces were cleaned for 30-60 min using an ultrasonicator, then dried in air. To produce the film of TiO2 NPs, the powder was first dispersed in TritonX100, acetic acid, and ethanol until it became a homogeneous paste nanomaterial. The paste was spread on the glass surface using a glass slide, and then the suspended powder was added as drops in the middle of the substrate and spread to form a thick film. The layer was then kept at room temperature for 30 minutes before being dried on a hot plate at 300 oC for 5 min [33].

2.3. Synthesis of ZnS/TiO2, CdS/TiO2, and CdS/ZnS/TiO2 nanocomposites as a hybrid structure:

The ZnS and CdS QDs act as light absorbers by the Successive Ionic Layer Adsorption and Reaction (SILAR) method. For the growth of ZnS QDs, the TiO2 semiconductor film was initially immersed in a 0.2 M Zn (CH3COO)2 aqueous solution for 2 min and then rinsed off with ethanol to remove excess ions and dried on a hot plate at 60 °C for 1 min. Then, the film was dipped into a 0.2 M Na2S aqueous solution for another 2 min to allow S2- to react with the preadsorbed Zn2+, leading to the formation of ZnS QDs. The loosely bound S2-ions were removed by rinsing the film in methanol, and then the film was dried at 60 °C for 1 min. The entire procedure is termed as one cycle, and the incorporated amount of ZnS can be increased by increasing the number of cycles to 3 cycles The films with different cycles were calcined at 300 °C for 5 min at a high temperature to increase the crystallinity of ZnS QDs [2]. The TiO2 semiconductor film was immersed in a 0.2 M Cd (NO3)2 aqueous solution for 2 min before being rinsed with ethanol and dried for 1 min on a hot plate at 60 oC. Then, the film was dipped into a 0.2 M Na2S aqueous solution for another 2 min to allow S2- to react with the pre-adsorbed Zn2+, leading to the formation of CdS QDs. The loosely bound S2- ions were removed by rinsing the film in methanol, and then the film was dried at 60 °C for 1 min on a hot plate at 60 oC. Then, the film was dipped into a 0.2 M Na2S aqueous solution for another 2 min to allow S2- to react with the pre-adsorbed Zn2+, leading to the formation of CdS QDs. The loosely bound S2-ions were removed by rinsing the film in methanol, and then the film was dried at 60 °C for 1 min. The entire procedure is termed as one cycle, and the incorporated amount of CdS can be increased by increasing the number of cycles to 3 cycles.

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To increase the crystallinity of CdS QDs, the films with different cycles were calcined at 300 oC for 5 min [7]. To grow a ZnS/CdS hybrid structure on TiO2, immerse the TiO2 film in a 0.2 M Zn(CH3CO)2 aqueous solution for 2 min, then rinse with ethanol to remove excess ions before drying for 1 min on a hot plate at 60 oC. Then, the film was dipped into a 0.2 M Na2S aqueous solution for another 2 min to allow S2- to react with the pre-adsorbed Zn2+, leading to the formation of ZnS QDs. This was followed by immersing the film of TiO2/ZnS into a 0.2 M Cd (NO3)2 aqueous solution for 2 min, rinsed off with ethanol to remove excess ions, and dried on a hot plate at 60 °C for 1 min. Then, the film was dipped into a 0.2 M Na2S aqueous solution for a nother 2 min to allow S2- to react with the pre-adsorbed Zn2+, leading to the pre-adsorbed Zn2+, leading to the formation of ZnS QDs. This was followed by immersing the film of TiO2/ZnS into a 0.2 M Cd (NO3)2 aqueous solution for 2 min, rinsed off with ethanol to remove excess ions, and dried on a hot plate at 60 °C for 1 min. Then, the film was dipped into a 0.2 M Na2S aqueous solution for another 2 min to allow S2- to react with the pre-adsorbed Zn2+, leading to the formation of CdS/ZnS hybrid QDs [7].



Schematic diagram of quantum dot deposition by SILAR Method

2.4. Methods of Analysis:

Several techniques were used to characterize the surface of prepared materials. X-ray diffraction patterns were reported using a Pan Analytical Model X 'Pert Pro, which was fitted with CuK α radiation ($\alpha = 0.1542$ nm), Ni-filter, and general area detector. A 40 kV accelerating voltage and a 40 mA emission current were utilized. In the 20 from 0.5–700 range, the diffract grams were registered. The Fourier transform infrared spectroscopy (FT-IR) of the prepared samples was measured using the KBr technique adopted by the Nicolet Is-10 FT-IR spectrophotometer (Thermo Fisher Scientific). The KBr technique was conducted roughly in a quantitative manner for all samples, since the sample weight and that of KBr were both held constant. A Field Emission Scanning Electron Microscope (FE-SEM) is used to examine the material's composition, particle size, and shape. For the samples prepared, a 30 kV acceleration voltage operating on a JSM-7500F electron microscope was reported. The optical absorption spectra of the samples were analyzed using Ultraviolet-Visible absorption spectroscopy (Spectro UV-Vis 2800, United States).

III. RESULTS

XRD analysis was applied for the determination of purity, crystallinity, and crystalline size. The XRD pattern spectra of the TiO2 NPs calcined at 400oC is depicted in Figure 1 (a). The XRD spectra of three samples show that the anatase phases are formed with the sharp peaks for TiO2 NPs. All diffraction peaks for the sample were congruent with the tetragonal structure of TiO2 NPs anatase phase and all diffraction peaks were in good agreement with JCPDS No. 21-1272. The diffraction angle peaks existing at $2\theta = 25.210$, 37.600, 47.940, 51.450, 53.920, 55.010, 62.590 and 65.420 correspond to the (101), (004), (200), (105), (211), (204), (116), and (215) lattice planes respectively. As shown in Fig. 1 (b), the pattern of the cubic phase of CdS, with preferred

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orientation along the (111) direction at $2\theta = 26.66^{\circ}$ [32]. This preferred orientation is frequently reported in SILAR-prepared CdS thin films [1, 6]. A small contribution of the planes of cubic CdS at $2\theta = 44.03^{\circ}$ is also observed [7]. Both XRD-peaks were indexed in the crystallography open database entry COD-1011251. The CdS sample exhibits with minor intensity the (100) at $2\theta = 24.83^{\circ}$ and (1 1 1) at 28.22° reflections of the hexagonal CdS phase (COD-1011054). A cubic structure of zinc sulphide QDs was identified from the XRD patterns in Fig. 1 (c) in which the peak positions of ZnS (JCPDS 65-9585) are shifted slightly too large as compared with the TiO2 NPs and CdS QDs [18]. The Sherrer equation was used to calculate the average crystallite size, which is as follows:

 $D = K\lambda/\beta \cos\theta$

(1)

Where: D is the crystallite size; K is a dimensionless shape factor, has a typical value of about

0.9 nm, λ =1.54A° is the X-ray wavelength; β is the broadening full width at half the maximum intensity (FWHM); and θ is the Bragg's diffraction angle [36].

The crystallite size of the pure TiO2 NPs, CdS QDs, and ZnS QDs nanostructures is shown in the below Table1. **Table 1: Crystallite Size, Thickness and Eg of pure nanostructures.**

Nanostructures	Crystallite size (nm)	Eg (eV)	Thickness (cm)
TiO ₂	11.80	2.39	0.0051
ZnS	25.91	2.25	0.01
CdS	10.3	2.2	0.013

Figure 2 shows the FTIR spectra curves of (a)-TiO2 NPs, (b)-CdS QDs, and (c)-ZnS QDs.

Figure 2(a) shows the FTIR spectra for sample TiO2 NPs calcined at 400oC and shows that the absorption band observed in the ranges of 400 confirms to 800 cm-1 is associated to the bending vibration (O-Ti-O) bonds in the TiO2 lattice [36-37], which indicates the presence of TiO2 as a crystalline phase. The sharp peak at 1620 cm-1 refers to the characteristic bending vibration of the OH group [38-39]. A broad absorption peak observed in the range of 3200 to 3800 cm-1 is due to the interaction of the hydroxyl group of water molecules with the TiO2 surface [36-37]. Fig. 2 (b) shows a very small related peak because CdS possesses a broad absorption band in the far infrared region. Also, Fig. 2(c) shows the sharp peaks, functional groups, and absorption peaks of ZnS QDs. The formation of ZnS QDs is confirmed by it. Figure 3 shows the FE-SEM images for the TiO2 QDs, CdS QDs, and ZnS QDs. From FE-SEM images of samples, Fig.3 (a), it is found the particles for TiO2 QDs are homogeneous and almost irregular in shape. It is found that particle size is increased and is highly agglomerated at 11.8 nm. Fig.3 (b) shows the micrographs of ZnS QDs, ZnS QDs are spherical shapes with a uniform of about 93.80 nm, and Fig.3 (c) shows that CdS QDs morphology and indicates an average size of about 14.21 nm.

Figure 4 shows the optical absorption spectra of UV–vis for TiO2 NPs, TiO2/ZnS NCs, TiO2/CdS NCs and TiO2/CdS/ZnS NCs respectively. The spectrum of the quantum dots deposition is different than the spectrum of TiO2 NPs, where the optical absorbance of TiO2/CdS/ZnS NCs is higher than TiO2 NPs, TiO2/ZnS NCs and TiO2/CdS NCs. which confirmed that the TiO2/CdS/ZnS NCs have a maximum absorbance.

To know the optical band gap, we have made a detailed calculation of the band gap using the Tauc formula; $(\alpha hv) = A$ (hv - Eg)n, here, α is the absorption coefficient, hv is the photon energy, and Eg is the energy gap, and n indicates the quality of the transitions. The term n is taken to be 2 for a direct transition and ½ for an indirect transition. The optical bandgap energy is investigated by plotting (αhv) 1/n versus photon energy and drawing the tangent to the curve that intersects with the energy axis at $\alpha = 0$. Fig.5 shows the Tauc plot of TiO2 NPs, TiO2/ZnS NCs, TiO2/CdS NCs and TiO2/CdS/ZnS NCs for direct transition and the estimated energy gaps of TiO2 NPs, TiO2/ZnS NCs, TiO2/CdS NCs and TiO2/CdS/ZnS NCs are 2.39 eV, 2.30 eV, 1.98 eV and 1.68 eV,

III. RESULTS



Figure1: XRD spectra of (a)-TiO2 NPs, (b)-CdS QDs and (c)-ZnS QDs.



Figure2: FTIR spectra of (a)-TiO₂ NP_s, (b)-CdS QD_s and (c)-ZnS QD_s.



Figure 3: FE-SEM of nanostructures (a)-TiO2 NP,, (b)-ZnS QD, and (C)-CdS QD,.

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Figure 4: Absorbance spectra of the TiO2 NPs, TiO2/ZnS NCs, TiO2/CdS NCs and TiO2/CdS/ZnS NCs.



Figure 5: Optical band gap of the TiO2 NPs, TiO2/ZnS NCs, TiO2/CdS NCs and TiO2/CdS/ZnS NCs.

IV. CONCLUSION

A TiO₂ nanoparticle was successfully synthesized by a sol-gel method. The TiO₂/ZnS, TiO₂/CdS, and TiO₂/CdS/ZnS nanocomposites were successfully synthesized by the SILAR method. We found that the particle size of CdS QDs is smaller than TiO₂ NPs and ZnS QDs and the absorbance of TiO2 NPs was increasing by quantum dot deposition under UV-irradiation (TiO₂/CdS/ZnS< TiO₂/CdS <TiO₂/ZnS <TiO₂ NPs(. Optical band gap energy in direct allowed transition for TiO2 NPs was decreased by adsorption of different quantum dots on the surface of TiO₂ to form hybrid structures where (Eg) for TiO₂/CdS/ZnS is lower than Eg of TiO₂/CdS NCs, TiO₂/ZnS, and TiO₂NPs. This means that deposition of hybrid quantum dots is the

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most efficient nanocomposites as a semiconducting nanomaterial and can be used as a photoanode in quantum dot sensitized solar cells (QDSSCs).

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